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AFRPL-TR-66-116

(UNCLASSIFIED TITLE)

EVALUATION OF COLUMBIUM CARBIDE NOZZLES FOR SOLID PROPELLANT ROCKET MOTORS

E. L. Olcott Atlantic Research Corporation

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FOREWORD

This report was prepared by Atlantic Research Corporation to document work performed under Air Force Contract AF 04(611)-8009. The contract called for the design and fabrication of two replicate columbium carbide nozzles by Atlantic Research Corporation. The nozzles were to be tested at the Air Force Rocket Propulsion Laboratory, Edwards, California. The nozzle design and fabrication took place between 6 February 1962 and 16 September 1962. Due to unforeseen difficulties in the development of the liquid simulator test device at the Rocket Propulsion Laboratory, testing of the second nozzle was not completed until February 25, 1965.

This program was administrated under the direction of the Air Force Rocket Propulsion Laboratory, Edwards, California, with Capt. D. England, Lt. R. Maxwell and Lt. E. Schneider serving as Air Force Project Officers. Mr. Eugene L. Olcott, Director of Materials Division, Atlantic Research Corporation, was responsible for the administrative and technical aspects of the program. Mr. Stanley Miller performed most of the work.

This technical report has been reviewed and is approved.

Earl M. Schneider Lieutenant, USAF

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1.0 INTRODUCTION

Refractory carbides possess the highest melting points of known refractory materials and, therefore, offer considerable promise for application under severe high-temperature duty cycles. The usefulness of the refractory carbides for rocket nozzle inserts has been limited by their relatively low thermal shock resistance. During the initial portion of the rocket nozzle firing cycle, a severe thermal gradient exists between the gas side and the back side of the nozzle insert, and as a result high tensile stresses are imparted to the backside. The relatively low fracture resistance of the refractory carbides often permits complete shattering of the carbide material from the thermal stresses. Researchers have labored to overcome the shortcomings of refractory carbides through such techniques as pre-stressing, grain size control, reinforcement, and dispersion of a low modulus phase. The Carborundum Company has had notable success in the use of a dispersed phase of graphite in a refractory carbide matrix to improve the fracture resistance of these materials. It was also found in subscale rocket nozzle tests that a thin layer of pure carbide could be used in connection with the carbide-dispersed graphite backup for acceptable fracture resistance. A discussion of the development of these materials may be found in Reference 1 and a discussion of the behavior of subscale nozzles made of such materials has been given in Reference 2. Recently, another approach to this problem was accomplished by the addition of sufficient carbon to a molten carbide to permit the precipitation of graphite flakes in the matrix during solidification. In this case, the low modulus dispersion is more in the form of graphite flakes.

Previous subscale tests on refractory carbides contining a pure carbide surface with a carbide-graphite backup had shown acceptable erosion and fracture behavior (Reference 2). The objective of this program was to scale up the test conditions to a 5,000-lb. thrust motor to determine the usefulness of refractory carbide nozzle inserts. The contract provided for the design and fabrication of two replicate nozzles to fit the Edwards test device which at the time of the contract initiation was a liquid simulator. Columbium carbide was selected as the carbide to be used in the nozzles because previous experience had shown this carbide to be sufficiently erosion resistant in the environment contemplated, and this carbide possessed a density less than that of many of the other carbides and would be advantageous for flight weight applications.

2.0 NOZZLE DESIGN

A. SERVICE CONDITIONS

The service conditions under which the nozzles were to be tested were initially specified as follows:

- 1. Thrust 5000 lbs.
- 2. Burn time 60 seconds (minimum)
- 3. Flame Temperature 5500°F to 6000°F
- 4. Chamber Pressure 500 psi
- 5. Nozzle contoured to provide for expansion to atmospheric exhaust pressure of 13.1 psi (Edwards AFB Rocket Site).
- 6. The propellant was later defined in more detail as follows:

$$I_{sp}$$
 = 264.56 sec $\frac{(1000)}{14.7}$
Density = 0.0635 lb/in³
 I_{flame} = 3185°K (5275°F)

Propellant Composition:

A1 = 20%

$$CH_2$$
 = 15%
 NH_4C10_4 = 65%
100%

The oxidation ratio calculated from the combustion products of the above propellant was 1.01%, which was sufficiently low to indicate an anticipated erosion rate of from .1 to .2 mil/sec for columbium carbide based on previous subscale tests.

The test nozzles were designed, fabricated and delivered prior to the change in test conditions which was occasioned by difficulties of the Rocket Propulsion Laboratory with the liquid test simulator. It was subsequently decided by the Rocket Propulsion Laboratory Personnel that the two nozzles would be tested in the Char Motor with the test propellant LPC-556 polycarbutene which has a calculated flame temperature of 5640°F (1000 psi motor pressure) and a calculated oxidation ratio of 1.26. The combustion products of this propellant are shown in Table 1. The maximum pressures obtained with the nozzle diameters selected on the basis of the liquid simulator turned out to be approximately 700 psi. It may be noted that the solid propellant Char Motor test conditions were more severe with regard to pressure, oxidation ratio and propellant temperature than

TABLE 1

CHARACTERISTICS OF TEST PROPELLANT, LPC 556

Composition: 68% ammonium perchlorate

17% aluminum

Balance, polycarbutene binder

T°F 5643

(Chamber 1000 psi)

I theo, shifting, 1bf-sec/1bm 263 I $_{\mathrm{sp}}^{15}$ standard 245, estimated

Composition of Principal Exhaust Products

Chamber	Shifting Exhaust
0.0498	0.0654
1.1629	1.2371
0.0355	0.0003
0.0218	0.0002
0.0162	0.0000
0.0011	0.0029
0.4967	0.4273
0.1129	0.0051
0.4588	0.5762
0.2884	0.0000
0.0303	0.0018
0.9144	0.8995
0.2975	0.2982
	0.0498 1.1629 0.0355 0.0218 0.0162 0.0011 0.4967 0.1129 0.4588 0.2884 0.0303 0.9144

the propellant initially selected for use in the liquid simulator. Fortunately, however, the more severe test conditions gave information more applicable to current propellant systems than the initial test propellant selection.

B. DESIGN

The columbium carbide insert design is shown in Figure 1. It features a reasonably symmetrical cross-section to minimize stress effects related to changes in section. The configuration provided a port/throat ratio of 1.18 at both entrance and exit ends. Retention is both by bearing at the exit end and by shear with a 7 degree ramp angle. Redundant retention was provided to permit retention of separate segments in the event that complete fracture occurred during the test firing. The insert was fabricated to provide maximum carbide density on the gas side and sufficient dispersion of graphite particles through the backup surface to provide the degree of fracture resistance believed to be necessary.

This insert was incorporated in the layout design shown in Figure 2. A preliminary version of this nozzle was thermally analyzed (Appendix I) and stress analyzed (Appendix II). The heat transfer calculations indicated the desirability of incorporating the graphite heat sink (Part 5 of Figure 2) to reduce the temperature of the hot side of the cast zirconia insulation member (Part 4 of Figure 2). This precaution was necessary only if the upper limit of propellant temperature required in the contract (6000°F) was used in the test program.

The castable zirconia insulation materials used for Parts 4 and 9 of Figure 2 were developed at Atlantic Research for use as a castable insulating material to provide a refractory backup for the hotter gas side materials. The development of this material is discussed in Reference 2. This material has the capability of being cast around brittle materials to offer firm backup support without precision machined fit-up. At temperatures above 4000°F this material softens and must be used with caution. The graphite segments used on each side of the throat insert can be considered a routine application for such materials. The phenolic-asbestos entrance material was designed to project into the liquid simulator thrust chamber over the water-cooled flange attadiment which was part of the simulator. The flange of the nozzle assembly was expected to, in turn, be cooled in part by the water-cooled flange to which it was attached. The phenolic-asbestos exit section was selected to be compatible with the propellant initially selected.

C. ENTRANCE MODIFICATION

After it became evident that the liquid simulator would not be available to test the nozzles, the Rocket Propulsion Laboratory

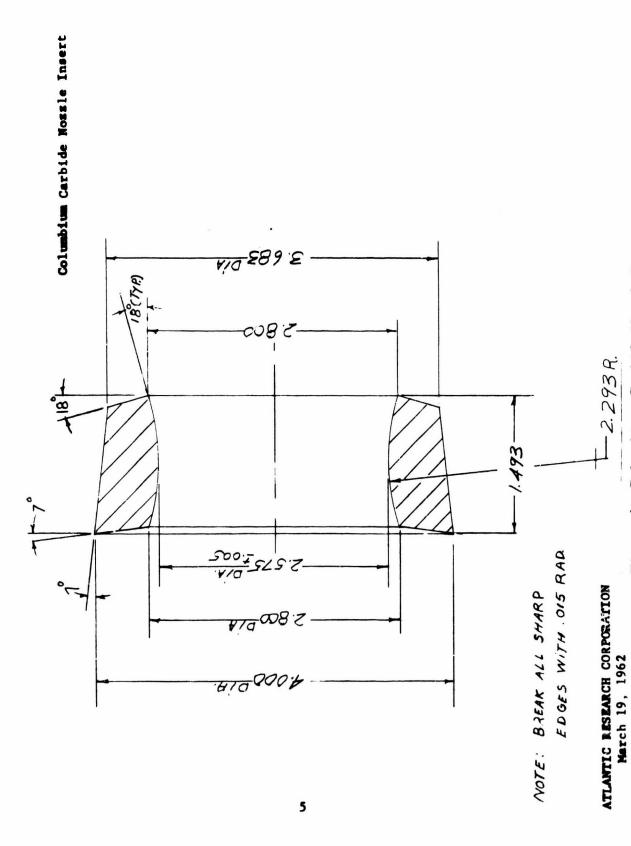
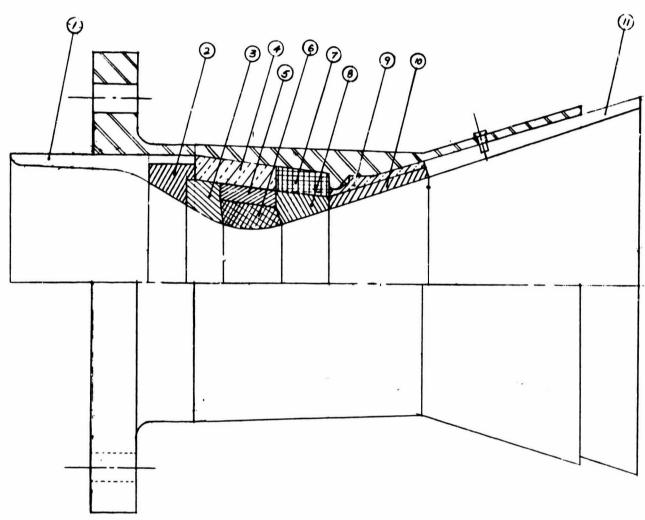


Figure 1. Columbium Carbide Nozzle Insert.



NO .	PART	MATERIAL
1	APPROACH INSULATION	PHENOLIC -ASSESTOS
2	APPROACH SEGMENT	ATS OR ANDG GRAPHITE
3	APPROACH SESMENT	ATJ OR ANDS GRAPHITE
4	INSERT BACK-UP INSULATION	CASTABLE ZP 02 CERAMIC
5	ENSERT HEAT SINK	ATJ DRAPHITE
6	ENSERT	COC + G RAPHITE DISPERSION
7	INSERT BACK-UP INSULATION + PETAINER	MACHINED GA CARBON
8	EXPANSION CONE SEG MENT	ATJ GRAPHITE
9	EXPANSION INSULATION	CASTABLE 27 0, CERAMIC
10	ERPANSION CONE SEGMENT .	ATT OR ATL GRAPHITE
11	EXPANSION CONE	PHE NOLIC - ASBISTOS

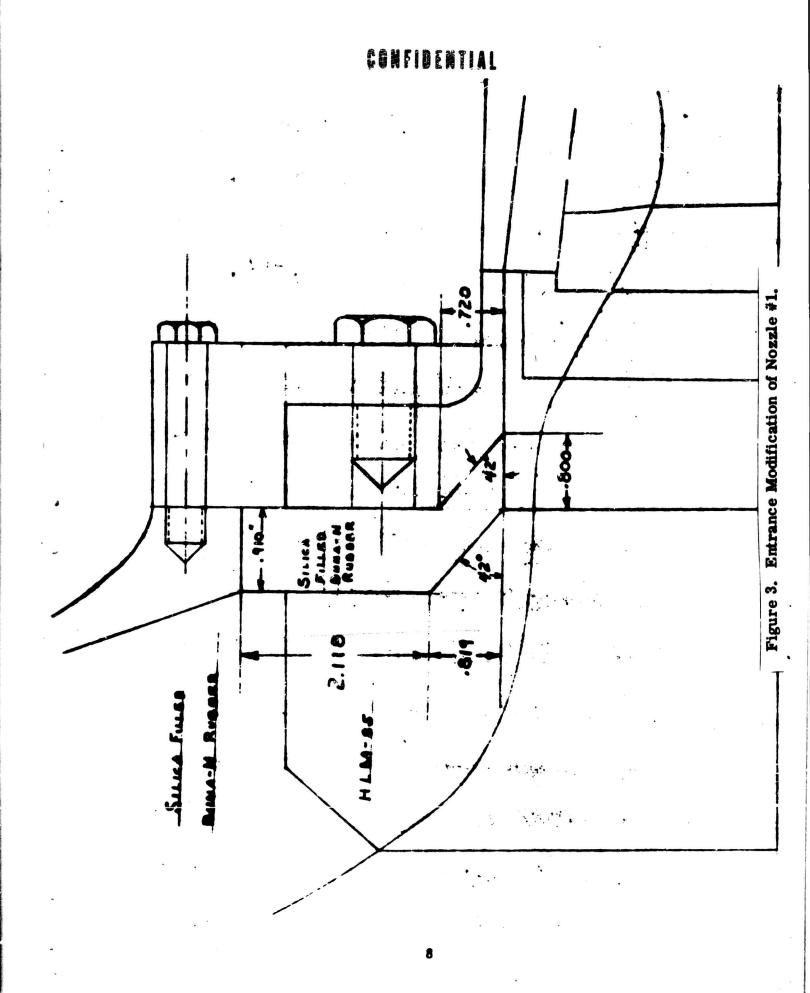
Figure 2. Columbium Carbide Nozzle Assembly.

Personnel designed and fabricated a transition flange to fit the closure of the solid propellant motor. This flange is shown in Figure 3 along with the mating portion of the columbium carbide nozzle. It was recognized at the time that the use of the silica filled Buna N facing the nozzle flange substituted for the water-cooled metal flange called for in the initial design arrangement was somewhat questionable for dependable performance, but was employed as an expedient. The transition adaptor for the second nozzle was also designed and fabricated by Rocket Propulsion Laboratory Personnel and is shown in Figure 4 (Drawing RB64D855). This design featured a more erosion resistant approach section which mated with the graphite entrance portion of the test nozzle.

D. FABRICATION

The two test nozzles were fabricated concurrently in the shops of Atlantic Research Corporation. The columbium carbide inserts were made by the Carborundum Company, Niagara Falls, New York. A dense columbium carbide layer 1/8 inch in thickness was backed up by a mixture of 30 volume per cent columbium carbide-70 volume per cent graphite dispersion. This carbide-graphite ratio was found to be fracture resistant from previous subscale tests. The inserts were fabricated by hot pressing at 2450°C and 6000 psi. Nozzle fabrication was completed on 16 September 1962.

The instrumentation on the nozzles consisted of two biaxial strain gauges located on the nozzle shell structure and four thermocouples located at various depths in the nozzle, one of which went through to the cold side of the carbide insert. During the re-work of the nozzle entrance section of the second nozzle at the Rocket Propulsion Laboratory all of the instrumentation was removed prior to firing. Figure 5 shows a completed nozzle prior to shipment.



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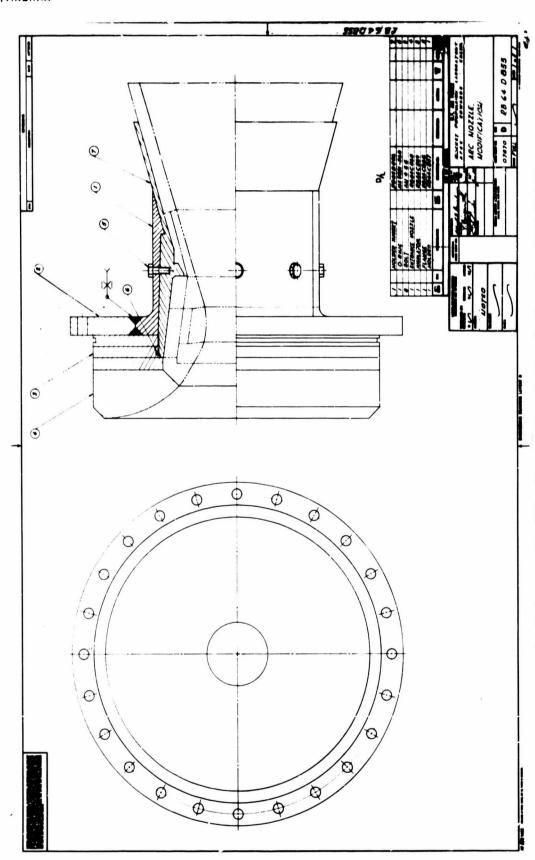
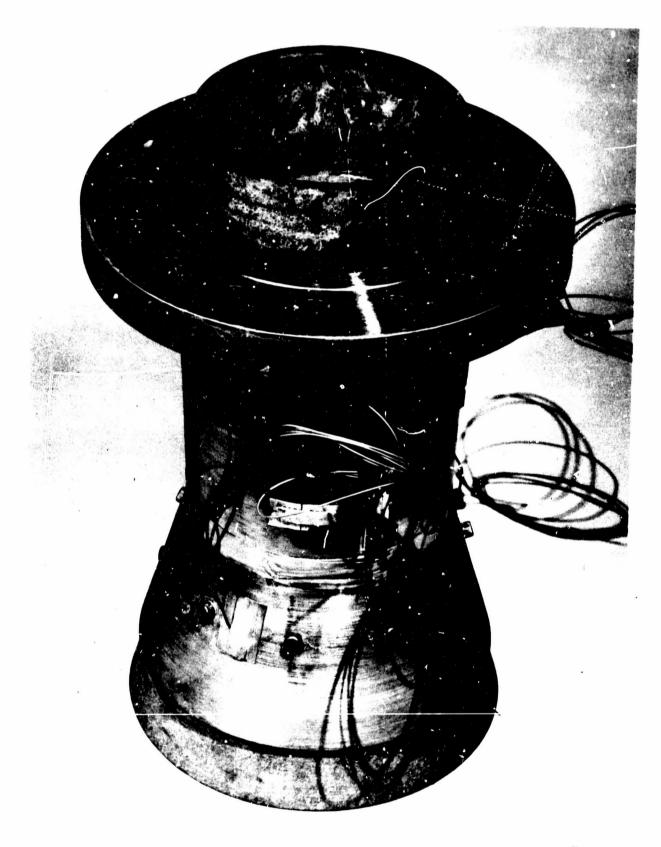


Figure 4. Entrance Modification of Nozzle No. 2, Drawing RB64D855.



A11332

Figure 5. Completed Nozzle for Attachment to Liquid Simulator.

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3.0 NOZZLE TEST RESULTS

The first firing, Nozzle E-1, was conducted on 18 March 1364, and the second firing, E-2, was conducted on 25 February 1965, at the Rocket Propulsion Laboratory. The mass flow of the Char Motor was 21 pounds per second. Firing data on the nozzles are shown in Table 2.

Firing E-1 was normal until 41 seconds at which time the movie film showed a gas leak occurred at the junction of the transition section at the base of the nozzle. The pressure-time curve of this firing is shown in Figure 6. It is believed that the silica filled Buna N insulation section and the rather thin phenolic-asbestos section which was designed to cover a water-cooled, stainless flange eroded in such a manner as to permit heating the steel at the flange section. A radial crack in the graphite entrance section opened to permit gas flow in this area. The gas leak rapidly increased and impinged upon the outside of the expansion cone and burned a hole in it as can be seen in Figure 7. The test insert removed from the nozzle assembly after firing is shown in Figure 8. The carbide insert suffered no discernable ill effects from the nozzle burn-through. The globule of aluminum oxide that can be observed in the entrance section was related to the low pressure burning after burn-through.

Firing E-2 progressed satisfactorily insofar as the insert was concerned for the full duration, 63 seconds. The pressure-time curve is shown in Figure 9. Figure 10 shows the second nozzle mounted on the Char Motor prior to test. Film coverage shows that at 38 seconds the aft end of the expansion cone started to deteriorate and was lost at 47 seconds. Figure 11 shows the nozzle after test. Figure 12 shows the removed insert after test.

A. NOZZLE INSEKT PERFORMANCE

The erosion rate for the first nozzle was calculated as 2.2 mil/sec by dividing the total radial change by the time to nozzle burn-through. Since some of the total erosion probably occurred during the period of low pressure burning, the actual erosion rate for the normal part of the firing cycle should be less than this value. The shape of the pressure-time curve for this firing (Figure 6) indicates that much of the total erosion occurred during the first 12 seconds. This behavior indicates the likelihood that some of the 1/8-inch thick dense carbide surface spalled and was carried away by the gas stream. Both nozzles showed substantial areas where the dense carbide coating was completely gone. The underlying 30-70 carbide-graphite mixture appeared to possess fair erosion resistance and was not unduly eroded. The second nozzle showed an average erosion rate of 1.6 mil/sec for the full duration. These erosion rates can be compared with a measured race of 2.2 mil/sec determined for ATJ graphite under similar firing conditions in another program (Reference 3).

TABLE 2

FIRING DATA FOR COLUMBIUM CARBIDE NOZZLES

Calculated Avg. Erosion Rate (mil/sec)	Nozzle Insert Material	<2.2	1.6
of Throat	After (in)	$\frac{2.757}{2.882}$	$\frac{2.785}{2.810}$
Diameter of Throat	Before After (in)	2.575	2.575
Average	Pressure (P) (psi)	553	-662-
Maximum	Pressure (psi)	685	769
Total Burning	Time (T) (Seconds)	54 ¹ 3005)	65
	iring	E-1 AFFTC 0005)	E-2 AFFIC 0018)

to burn through

 2 minimum and maximum dimensions

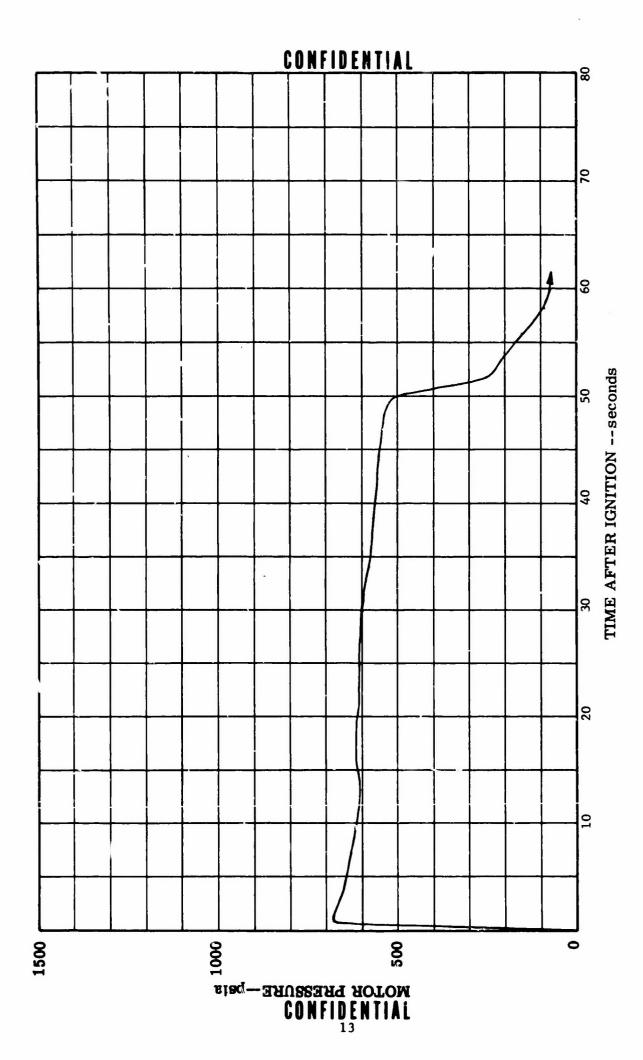


Figure 6. Pressure-Time Curve for Firing of 1st Nozzle.



Figure 7. Post-Firing View of Nozzle No. 1.

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Figure 8. Columbium Carbide Nozzle Insert After Firing No. 1.

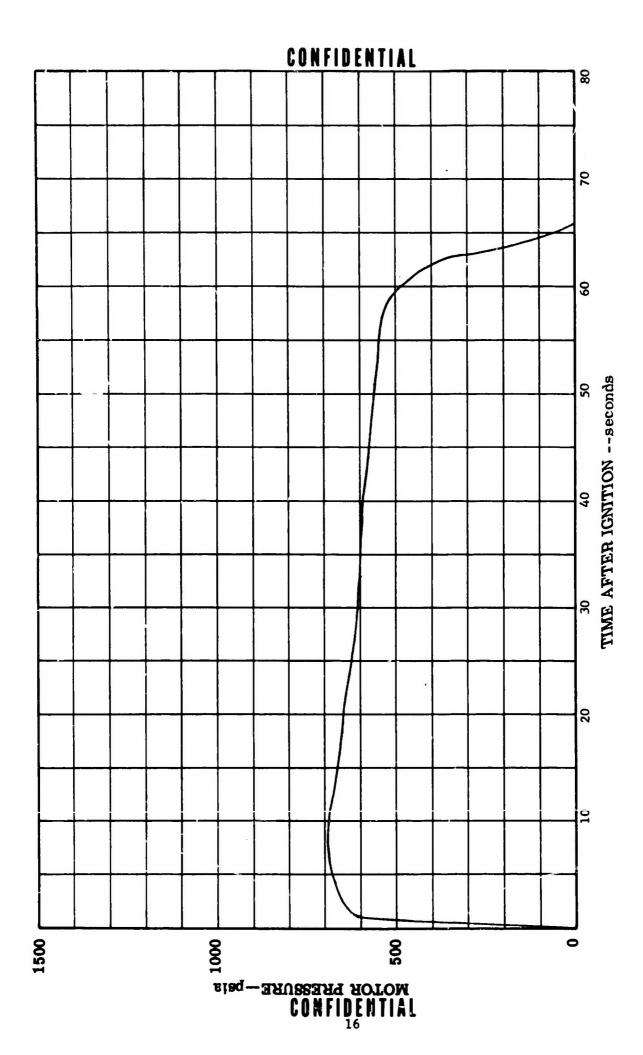


Figure 9. Pressure-Time Curve for Firing of Second Nozzle.

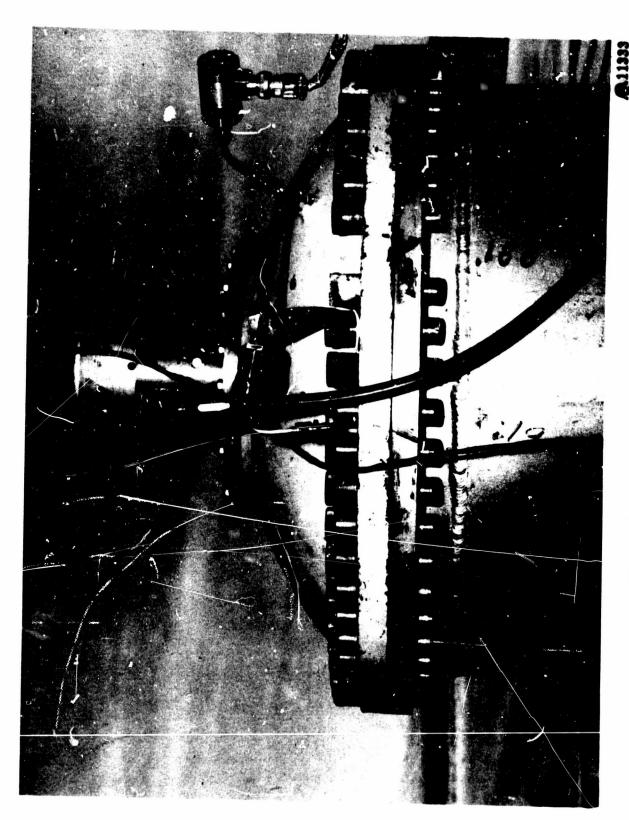


Figure 10. Second Nozzle Before Test.

Figure 11. Post-Firing View of Second Nozzle.



E-2

A11335

Figure 12. Columbium Carbide Insert After Test of Second Nozzle.

A few tight axial cracks were observed on the outside of the second insert. These cracks had the appearance of cool-down cracks. The first nozzle contained no visible outside cracks. The portions of the graphite-free carbide surface which remained on the gas side of the insert contained tight cracks both in the axial and transverse direction generally spaced at 1/2-inch intervals. These cracks appear to have been present during the firing but did not effect the performance or the erosion except as they may have permitted portions of the graphite-free carbide surface to be removed. If the bond strength of the carbide surface the carbide-graphite substrate were uniformly good, the cracks would not have caused any losses of the carbide surface. As can be seen in Figure 8 and 12, the carbide inserts were still in one piece after firing and showed no major structural deficiencies. The performance of such brittle materials under the severe duty cycle can be considered good.

Microscopic examination of the cross-section of the nozzle insert after Firing No. 2 showed no unusual deterioration. The carbide layer remaining was of high density, well bonded and of fine grain size. No significant reaction layer was observed on the gas side. The cracks which were visible on the gas side were tight and did not appear to be sources of accelerated erosion as can be observed in Figure 13. The portions of the gas side surface which contained no dense carbide coating blended to a section of a tapered thickness of dense carbide coating where the carbide relations can be observed in Figure 14. This appearance indicates that, in the sections examined any loss of the 1/8-inch thick carbide surface was gradual rather than by failure at the carbide to carbide-graphite interface.

The location of the composite substrate on the cross-section examined after fixing indicated that the nominal 1/8-inch thick carbide layer had not been attained at the entrance face edge of the insert. In both nozzles no dense carbide surface material remained in this location. There are indications that it may have originally been as little as 10 or 20 mils in thickness at this point. Although the influence of the thin coating at this location cannot be determined, it is likely that some of the erosion at the throat section could be attributed to this manufacturing irregularity.



A11337

Figure 13. Photomicrograph (X 60) of Etched Cross-Section of Nozzle No. 2.



A11338

Figure 14. Photomicrograph (X 60) of Etched Cross-Section of Nozzle No. 2 at Area Where Carbide Layer was Eroded.

1. Carbide Erosion Mechanism

Current and previous studies have shown that the principal erosion mechanism for a refractory carbide at the temperatures involved with the current Char Motor propellant is oxidative chemical attack (Ref. 2 and 4). The common oxidative combustion species, CO₂, and H₂O, have been found to be very reactive with hot-pressed carbides made by the same procedure utilized for the nozzle inserts. Under laboratory test conditions significant differences have been found in the oxidation effects on various refractory carbides which are caused by differences in the characteristics and protective quality of the metal oxide product produced at the carbide surface. For columbium carbide the protection by the oxide failed to be effective at a temperature well below 4000°F.

Work to date, although not complete, indicates that both hafnium carbide and zirconium carbide are more oxidation resistant (Ref. 4). The protective action of the oxidation product formed on hafnium carbide appears to persist up to temperatures near 5000°F under laboratory conditions. The presence of a metal oxide condensed phase, such as aluminum oxide, in the solid propellant exhaust will undoubtedly reduce the effectiveness of protection by an oxide layer, but it does seem likely that the relative performance of the various carbides may be related to that observed in laboratory tests. Thus, the investigation of other carbide materials for nozzle service might produce improved results.

B. OTHER NOZZLE COMPONENTS

Some of the other nozzle components did not perform so well. The graphite entrance piece which had been applied to the nozzle during the re-work operation at the Rocket Propulsion Laboratory to effect the fit in the solid propellant motor showed a gaping axial crack after each firing. The crack in the second nozzle is shown in Figure 15. of the crack and the local erosion which occurred urderneath the graphite indicated the open crack was present for much of the firing duration. The opening in the graphite entrance piece permitted gas impingement on the silica-Buna N insulation in the first nozzle, which subsequently resulted in burn-through at this point. In the second nozzle, the gas flowing behind the cracked graphite entrance piece impinged on the silica-phenolic material, the asbestos-phenolic insulation and also on the castable zirconia insulation backup. None of these materials are sufficiently resistant to gas impingement to remain unaffected and so a groove was formed which extended to a point behind the carbide insert itself. This penetration was not sufficient to significantly affect the performance of the nozzle by the end of firing, 63 seconds.

It was also indicated that the phenolic asbestos expansion cone was of insufficient thickness and support to withstand the full firing duration under the test conditions.



Figure 15. Entrance Section of Nozzle No. 2 After Firing.

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Figure 16 shows the temperature traces obtained from the first firing. The three thermocouples were located on the steel housing as shown. It is evident from the temperature increase measured by thermocouple T1 that this thermocouple was located near the region of nozzle burn-through which coincided with the axial crack in the graphite entry se section. Thermocouples T2 and T3 were essentially unheated up to the point at which instrumentation was lost.

Figure 17 shows the recording of strain gauges mounted on the nozzle barrel. Items 2 and 3 are a two strain gauge rosette located near the flange. The calculated principal stresses from these strain gauge readings are:

maximum normal stresses - 15,170 psi tensile minimum normal stress - 10,550 psi tensile maximum shear stress - 2,305 psi

Items 4 and 5 also form a two strain gauge rosette which is located on the nozzle barrel at a point opposite the insert retention shoulder. Calculations of principal stresses from these gauges are:

maximum normal stress - 21,600 psi compressive minimum normal stress - 16,800 psi compressive maximum shear - 2,305 psi

It should be noted that the strain gauges were not temperature compensated and so the readings used in the above calculations were taken 4 seconds after ignition. After this time the temperature effects influenced the gauge output, and further stress calculations could not be made.

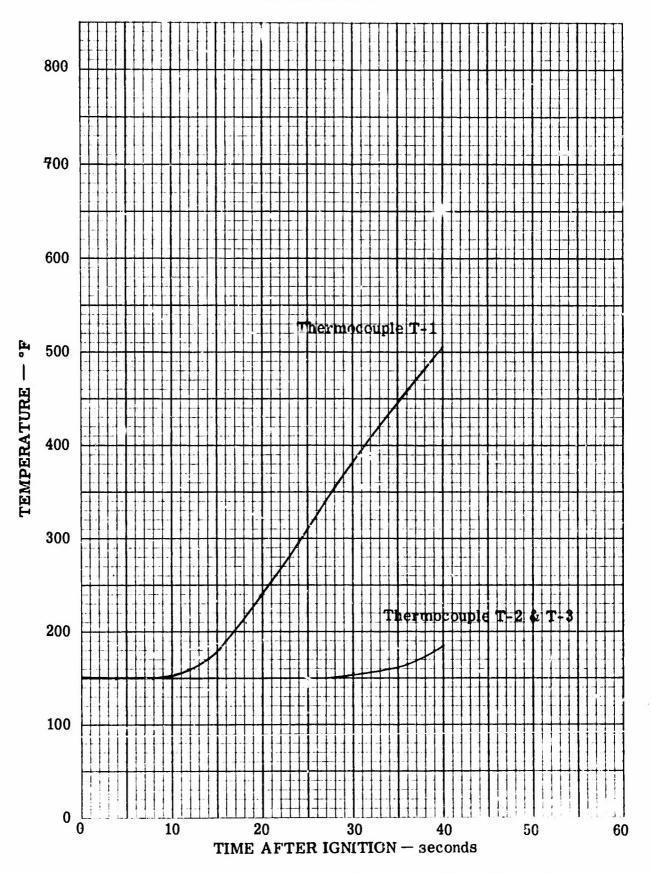


Figure 16. Temperature-Time Traces for Firing No. E-1.

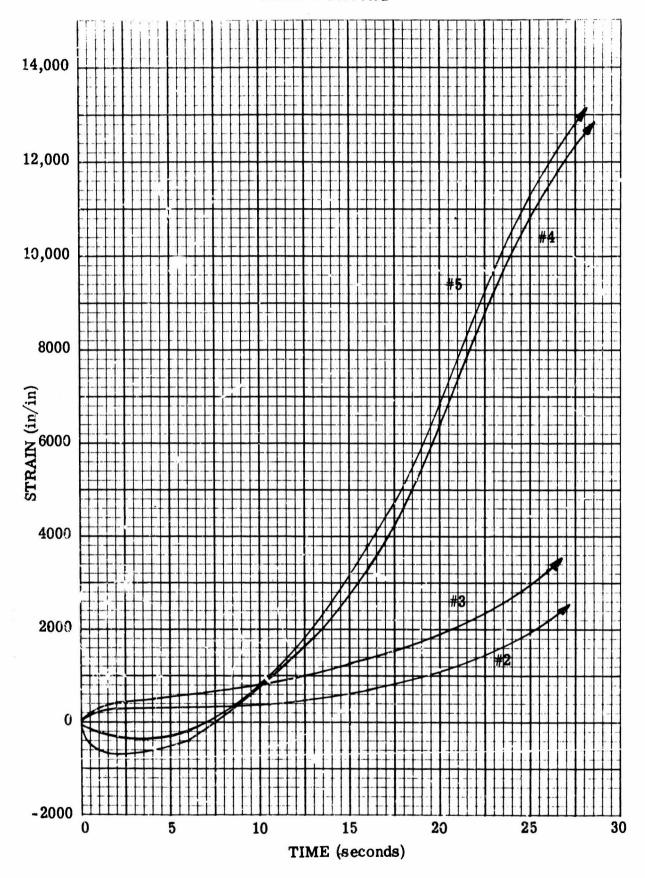


Figure 17. Recording of Strain Gauges, Firing No. E-1.

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4.0 CONCLUSIONS AND RECOMMENDATIONS

Columbium carbide nozzle inserts performed satisfactorily in the 5,000-lb. Char Motor with an oxidizing solid propellant. Erosion encountered was believed to be due primarily to oxidation of the carbide. No significant fracture of the brittle insert material occurred. Nozzle inlet damage related to a wide crack in the graphite entrance cap of each nozzle probably increased the nozzle insert heating and contributed to the erosion of the carbide. Advances in the state-of-the-art since the manufacture of the tested carbide inserts should permit modest improvements with similar current materials. In particular a manufacturing discrepancy which left a very thin carbide layer at the entrance face could be avoided.

Other work at Atlantic Research Corporation has indicated that other refractory carbides such as hafnium carbide are more suitable in oxidizing environments than is columbium carbide. Additional feasibility studies of such materials are believed desirable.

Because of the unusually high temperature compatibility of the refractory carbides, they should be considered for selected environments where they may be chemically compatible with a suitable environment.

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- 1. Carborundum Company, Quarterly Progress Report, Refractory Materials Suitable for Use in Guided Missile Propulsion Systems (U), Contract NOrd 18136, CONFILENTIAL.
- Special Projects Office Materials Program (U), NOw 62-0511, Final Report. April 1963, CONFIDENTIAL.
- Atlantic Research Corporation, <u>Castable Carbon for Nozzle Applications</u>, AFRPL TR-64-183, Third Quarterly Progress Report, Contract AF 04(611)-9718, December 1964.
- 4. Batchelor, J. D., Atlantic Research Corporation, Behavior of Nozzle Materials Under Extreme Rocket Motor Environments, Quarterly Progress Report, Contract NOw-0393-c, April 1965.

APPENDIX I

HEAT TRANSFER ANALYSIS OF COLUMBIUM CARBIDE

The heat transfer analysis of the columbium carbide nozzle is putlined in this Appendix. The enclosed plots show the transient temperature time history of the nozzle during the motor firing. It was assumed that the combustion temperature of the propellant gases was 6000°F. On this basis, the expected surface temperature of the nozzle throat hot side surface at 60 seconds will be about 5400°F. The cold side temperature will be about 4200°F. The surface temperature of the approach section at the end of 60 seconds of firing will be about 5200°F. The surface temperature of the exit portion of the nozzle insert was estimated to be about 150°F less than that of the approach section.

At the time the thermal analysis of the nozzle design was conducted, temperature dependent properties of the composite columbium carbide material were not available. For this reason, simplified analytical techniques believed to be adequate for the purpose intended were applied. Ordinarily, a two dimensional transient heat transfer program for the 7090 computer is used for problems of this type.

Because of the rework of the nozzles for use on the test motor selected, the temperature instrumentation was removed and correlation of measured temperatures with calculated temperatures could not be obtained.

HEAT TRANSFER CALCULATIONS FOR EDWARDS NO ZZLE

PHYSICAL PROPERTIES OF PROPELLANT GASES:

(according to K. Woodcock)

Mu= 27.0 1/2 mole

This equation was found to be accurate in numerous

experiment's by Aerojet, see

(1) St. Coluce, Experimental Determination of Solid Rocket

(1) Si. Colucc., "Experimental Determination of Solid Rocket

Nozzle Heat Transfer Coefficient," Proceedings of 5th

AFBMD-STL Acrospace Symposium, Vol. II Academic Press, 196

(2) SS Grover, "Analysis of Nozzle Heat Transfer Coefficient,

AGC TM 113 SRP 30 Apr 1959.

(3) E.M. Sadownick, "Investigation of Haterials Capabilities of

Materials Systems in Solid Rocket Hotors", Pait IT,

WADC TR 59-602, Part II

$$G = \frac{W}{\pi D^2/4}$$
 (2)

For propellant used

Cw = 0.0064 Lbm/66 . sec

and in motor

Pc = 500 psia Dt = 2.575 in

Converting all data into hours foot units, substituting in (4)

$$h = 0.023 \frac{23.1^{0.8} 72100^{0.8} 0.2145^{1.6}}{D^{1.8}} \frac{0.472^{0.4} 0.118^{0.6}}{0.211^{0.4}}$$
 (5)

when D is in inches, then

Because gas properties, cp, k, and μ vary with temp. in this modified Pr number in such a manner as not to affect this Prnumber by more than 5%, we can assume that this equation(1) is valid for entrance section, throat section and exit section heat transfer calculations.

In heat transfer calculations of nosale walls the adiabatic wall temperature is used as the temperature which is the driving potential. The adiabatic wall temperature is defined as

Taw = Nrs (Ts-Tm) + Tm

where Ts - is the "total" or "stagnation" tump Tm - is the moving-stream temperature and Nrs-is the vecovery factor (for turbulent boundary layer = Npr's; where Pr : Cpl

In the proposal we are told to assume as the total temperature is 5500-6000°F. We will use the higher value of 6000°F or 6460°R. The noving-stream gas temperature will be calculated from Ts-Tm = 12/2ge Jep (ideal gas) The Pr is equal to 0.472 × 5.862 × 10.5 3.3 × 10.5 or 0.839. The recovery factor is then 0.94

Position	A/A+	TWITS	Tm, or	47.	Taw, or	Taw, *F
। 2 3	1.183 1.000	.9635 .9091 .8223	6220 5,875	182 282 540		5985 5965 5836

Properties of Nozzle Material

Using NAVORD Report 5562. Temperature tables, Part 2, One-layer Cylindrical Shell, Internal Heating, One Space Variable, linear

for entrance, inside surface, no conduction past insert

$$\alpha_1 = \frac{v_i}{r} = .673$$

.06

.43

.461

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for entrance, outside surface r= ro h. = 0 d = ro-re = 2.0-1.4 = 1 × 0 = 0 x; = 15.2 a, = .673 طز (ط,=,٦) او 40 बं (४,=.6) ७ ४० ~ .01 ,002 .005 100, 810, 210, ,020 , 040 . 0 31 ککه، .02 ,185 .049 .100 .150 .110 .04 145 ,220 .310 449 . 163 , 284 , 387 .265 371 ,10

470

.688

Tw 2 70° + (5985-10) T;	. 21 . 21	6,24 12,48	Tw.°F 176 661
	104	24.96	1750
	,06	37.44	. 2665
	110	62.40	3917

.631

. 834

.414 ,439 ,561

.769

,627 ,651

For Nozzle Theory, Inside Surface & ourside surf.

10 = 1.920 in = .160 [1

11 = 1.287 in = .1072 [1

12 = 1.287 in = .1072 [1

13 = 0

14 = 0

15 = 0

16 = 0

16 = 0

17 = 0

17 = 0

17 = 0

17 = 0.671

17 = 0.671

17 = 0.671

17 = 0.671

17 = 0.671

17 = 0.671

4-		مر: (م	(,=,6)	५; ((4,2.7)	Twof		d=,671	h
~	t, sec	10	40	10	40	(w, r	10	17.6	40
0.01 0.04 0.06 0.10	5,75 11.5 23.0 34.5 57,5		.849 .886 915 933 .957	,638 ,729 ,792	.851 .889 .927 .951 .978	3810 4260 4690 4982 5375	,637 ,723 ,781	,774	.850 .888 .924 .946

OUTSIDE SURFACE

•	2	طز (درد. د)	K; (4,2.7)	Tw &	× . 671		
₹	t, see	x; (x,e.6)	10 40	10	126 40		
. 01 . 02	5.7.5 11.5	700, 400, 720, 1601	.020, 040 ,110 ,185	183 .019	,019 .030		
.04	23.0	1145 .220	.310 ,449		7,095,147		
.06	34.5 57.5	, 461 , 593	,470 .631 ,688 .834		9 .446 .556		

ATLANTIC RESEARCH CORPORATION ALEXANDRIA, VIRGINIA

ASSUME NOTILE THROAT HATERIAL HAS K = 20 BTU/hr-ft-OF

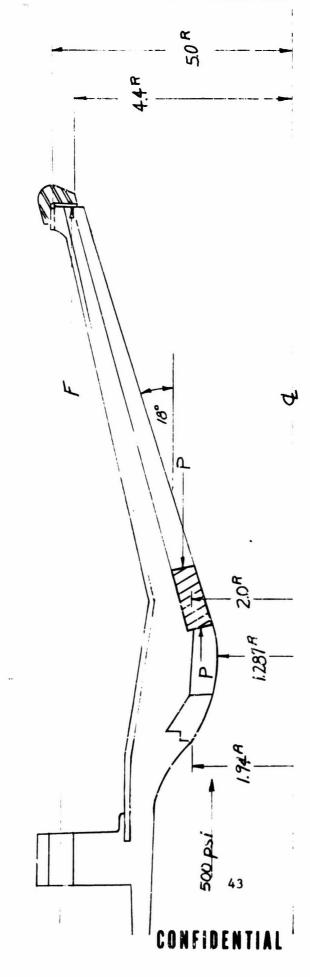
HOT SIDE	1, sec. 2,9 02 5,8 04 11.5 06 17.2 10 29 20 57.5	.388 .632 .470 .708 .524 .753 .609 .811	4 10 .314 ,552 .393 ,638 .488 .728 .567 .792 ,671 .877	4 8.8 10 313 .504 .551 391 .587 .636 .493 .675 .722 .552 .735 .781 .659 .820 .860 .815 .927 .952	3150 3572 4100 4410 4965 5606
	2.9 5.8 11.5 17.2 29.0 57.5	4. 10 .001 .002 .016 .031 .085 .146 .165 .764 .312 .464 .576 .752	.060 .110 .187 .310 .304 .470 .490 .688	4 8 10 .009 .014 .015 .047 .081 .090 .157 .269 .161 .380, 410 .440.588 .625 .710.837 .869	Tw. F 150 530 1485 2735 3577 5062

APPENDIX II

STRESS - ANALYSIS OF COMBUSTION

IMPULSE CARBIDE NOZZLE

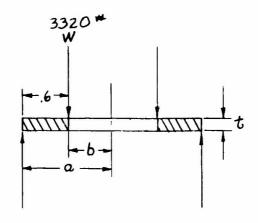
1



$$F = \frac{500 (1.94^2 - 1.287^2)}{2 \times 4.4} = 0.00 \#/1n$$

Total Load on Ring, $V^{\dagger} = 120 \times 8.8 \text{M} = 3320 \text{ }^{4}$

 $P = \frac{3320}{2\times2\pi} = 264 \#/\text{in Force on First Insert Downstream}$



$$\frac{a}{b}$$
 $\frac{5}{4.4}$ = 1.135
 β = 1.1 (-) (Tangential)

$$t^2 = \frac{1.1 \times 3.32}{25} = .146$$
.5000 psi
 $t = .382$ " (Conservative)

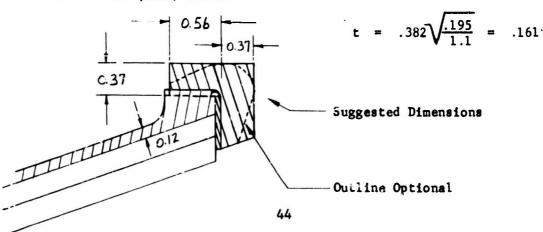
Alternate Way

(TIMOSHENKO II)
$$t^2 = \frac{6 \times 120 \times .6 \times 4.7}{25000 \times 4.4 \ln 1.135} = .146$$
 $t = .382''$

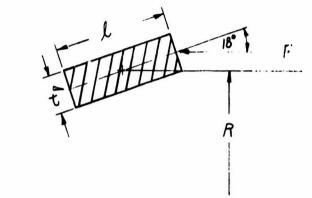
ROARK X-18



For Radial (max.) stress



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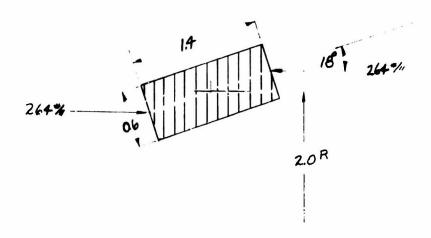


$$F = \frac{3320}{2K\Pi} = \frac{528}{K}$$

Approx. Bending Stress

$$\sigma_{\rm B} = \frac{12 \, {\rm FR} \, \underline{\ell}^2 \, \sin \alpha}{\underline{\ell}^3 {\rm t}} = \frac{528 \times 6}{\ell {\rm t}} \, \sin \, 18^{\circ} = \frac{983}{\ell {\rm t}}$$

This stress is highest in insert 1.



Approximate Bending Stress

$$\frac{6 \times 2 \times 264 \times 1.4 \sin 18}{1.4^2 \times .6} = \pm 1168$$

Approximate Compression

$$\frac{264}{.6} = -440$$

Maximum Stress

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